

REMARKS

Claims 65-72 remain in this application. Claims 1, 58-64 and 73-78 have been cancelled by this response. Claims 66-72 have been amended to depend from claim 65. The specification has been amended to remove extraneous or duplicate matter, improve organization and to correct clerical errors. These amendments are presented in a substitute specification, which has been provided in marked-up and clean forms in compliance with 37 C.F.R. 1.125(b) and (c). By these amendments, no new matter has been added.

Claims 1, 58-64 and 66-72 have been rejected under 35 U.S.C. 112, first paragraph. These rejections are respectfully traversed. Claims 66-72 have been amended to depend from claim 65, which is indicated as allowable. The remaining ones of the rejected claims have been cancelled. These rejections are therefore moot, and should be withdrawn.

The specification has been objected to as confusing. A substitute specification is provided, which clarifies the disclosed subject matter. In particular, Table 2 has been corrected so that it conforms to the text of the specification, and the error noted by the Examiner is corrected. The undersigned was able to discern the error and make the appropriate correction by applying a basic knowledge of chemical notation to the patent disclosure. In particular, in column 1, row 1 of Table 2, the first step is described as "Mixture of difunctional and trifunctional chlorosilanes reacted with a sub-stoichiometric amount of sodium (where $y < [3a+2b]$).". In addition, from the left side of the equation in column 2, row 1, "[3a+2b]" clearly refers to the stoichiometric amount of sodium needed to react with all the chlorine atoms. Therefore, "y" is clearly intended to denote some amount of sodium less than $3a+2b$; i.e., a sub-stoichiometric amount that should be added to the left side of equation 1. The former notation " $y/(3a+2b)$ " clearly makes no sense, as the Examiner pointed out. That the error was evident to the Examiner and that an appropriate correction was apparent to the undersigned supports the fact that

Serial No. 09/782,945
August 31, 2006
Page 6

this was merely a clerical error, the correction of which would have been readily apparent to one of ordinary skill. The correction therefore entails no new matter.

The remainder of the specification has been reviewed and certain minor corrections have been made, none of which should require further comment. This objection should therefore be withdrawn.

Applicants acknowledge with appreciation the allowance of claim 65. Claims 66-72 have been amended to depend from claim 65, and are therefore also allowable.

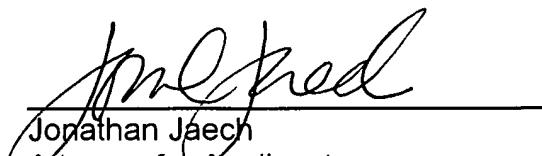
In view of the foregoing, the Applicants respectfully submit that claims 65-72 are in condition for allowance. Reconsideration and withdrawal of the rejections is respectfully requested, and a timely Notice of Allowability is solicited.

To the extent it would be helpful to place this application in condition for allowance, the Applicants encourage the Examiner to contact the undersigned counsel and conduct a telephonic interview.

The Commissioner is authorized to charge any shortage in fees due in connection with the filing of this paper, including extension of time fees, to Deposit Account No. 50-3683.

Respectfully submitted,

Date: August 31, 2006



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Enclosures: Substitute Specification (Marked-up)
Substitute Specification (Clean)

Serial No. 09/782,945
August 22, 2006
Page 7



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APPLICATIONS OF METHODS OF SYNTHESIZING PRECURSORS FOR PHOTOCURABLE PRE-CERAMIC POLYMERS

[0001] This application is a continuation-in-part of an application filed Jun. 3, 1999 under Ser. No. 09/325,524, now Patent No. 6,403,750, and is also a continuation-in-part of application Serial No. 09/680,828 filed Oct. 6, 2000, which applications are incorporated herein, in their entirety, by reference.

BACKGROUND OF THE INVENTION

[0002] The field of the invention is specific applications of The present invention concerns methods of synthesizing photo-curable preceramic polymers, and more particularly, poly(ethynyl)carbosilane. chemistry to specific applications and more particularly to specific applications such as impregnation and/or coating of ceramic filters for use in combustion engine exhaust systems.

[0003] U.S. Pat. No. 6,174,565 teaches an abrasion resistant ceramic material which includes a porous ceramic material having a surface and a preceramic polymer which is infiltrated into the surface. The preceramic polymer is converted to a ceramic state.

[0004] U.S. Pat. No. 6,165,240 teaches a filter apparatus which is for use in a forced air ventilation system. The filter apparatus has a first fan folded sheet of a washable first type of filter material, a second fan folded sheet of a second type of filter material, and fanfolds of the first and second fan folded sheets are substantially equally shaped and folded together such that the fanfolds of the first and second fan folded sheets are interdigitated to form a filter material assembly. The first type of filter material is preferably an electrostatically self charging washable material such as woven polypropylene and the second type of filter material is preferably disposable and contains activated charcoal. The filter material assembly may be removably disposed in a rectangular filter frame for mounting the filter material assembly in the forced air system.

~~The filter frame has at least one crank shaft rotatably disposed between two opposed side walls which are between an open front end and an open back end of the frame.~~

~~[0005] U.S. Pat. No. 4,816,497 teaches a method which is for the preparation of ceramic materials or articles by the pyrolysis of mixtures of vinyl or allyl containing preceramic silazane polymers, mercapto compounds, and photoinitiators. The mixtures are rendered infusible prior to pyrolysis by ultraviolet (UV) irradiation. This method is especially suited for the preparation of ceramic fibers.~~

~~[0006] U.S. Pat. No. 5,051,215 a rapid method of infusibilizing (curing) preceramic polymers which includes treatment the polymers with gaseous nitrogen dioxide. The infusibilized polymers may be pyrolyzed to temperatures in excess of about 800.degree. C. to yield ceramic materials with low oxygen content and, thus, good thermal stability. The methods are especially useful for the production of ceramic fibers and, more specifically, to the on-line production of ceramic fibers.~~

~~[0007] U.S. Pat. No. 6,063,327 teaches a method of making a shaped article and composite therefor which includes of a shaped tool, such as a mold and providing a composite of a quartz fabric disposed within a polysilazane which is moldable at a temperature below the curing temperature of the polysilazane and cured by catalyst. The polysilazane can be optionally filled with particulate material. A compaction pressure is applied to the composite to insure contact of the composite and the tool and prevent loss of less reactive polysilazane components. The composite is then cured to hardness by increasing the temperature applied to the composite at a rate of about 10.degree. C./minute to a temperature of about 150.degree. C. and holding that temperature of about 150.degree. C. up to about 4 hours to provide a cured polymer. The cured composite is removed from the tool. The polymer can then be converted to a preceramic by heating the cured polymer to a temperature sufficiently high in an inert atmosphere or ammonia gas for a sufficient time. The removal of carbon will occur when processed in ammonia. The preceramic can then be reprocessed by repeating the above procedure after cooling and reinfiltating with uncured resin.~~

[0008] U.S. Pat. No. 4,806,612 teaches pre-ceramic acetylenic polysilanes which contain $-\text{CH}(2)(\text{w})\text{C.tbd.CR}'$ groups attached to silicon where w is an integer from 0 to 3 and where R' is hydrogen, an alkyl radical containing 1 to 6 carbon atoms, a phenyl radical, or an $-\text{SiR}''(3)$ radical wherein R'' is an alkyl radical containing 1 to 4 carbon atoms. The acetylenic polysilanes are prepared by reacting chlorine or bromine containing polysilanes with either a Grignard reagent of general formula $\text{R}'\text{C.tbd.C(CH}(2)(\text{w})\text{MgX}'$ where w is an integer from 0 to 3 and X' is chlorine, bromine, or iodine or an organolithium compound of general formula $\text{R}'\text{C.tbd.C(CH}(2)(\text{w})\text{Li}$ where w is an integer from 0 to 3. The acetylenic polysilanes can be converted to ceramic materials by pyrolysis at elevated temperatures under an inert atmosphere.

[0009] U.S. Pat. No. 4,800,211 teaches 3-Hydroxybenzo[b]thiophene-2-carboxamide derivatives which have been prepared by: (1) treating a substituted 2-halobenzoate with a thioacetamide; (2) treating a substituted thiosalicylate with an appropriately substituted haloacetamide; and (3) further synthetic modification of compounds prepared above. These compounds have been found to be effective inhibitors of both cyclooxygenase and lipoxygenase and thereby useful in the treatment of pain, fever, inflammation, arthritic conditions, asthma, allergic disorders, skin diseases, cardiovascular disorders, psoriasis, inflammatory bowel disease, glaucoma or other prostaglandins and/or leukotriene mediated diseases.

[0010] U.S. Pat. No. 4,588,832 teaches a novel and economical route for the synthetic preparation of a 1-alkynyl trihydrocarbyl silane compound. The method includes the steps of reacting metallic sodium with a hydrocarbyl substituted acetylene or allene compound to form a substituted sodium acetylide and reacting the acetylide with a trihydrocarbyl monohalogenosilane in the reaction mixture which is admixed with a polar organic solvent such as dimethylformamide.

[0011] U.S. Pat. No. 4,505,726 teaches an exhaust gas cleaning device provided with a filter member which collects carbon particulates in exhaust gases discharged from a

~~diesel engine and an electric heater for burning off the particulates collected by the filter member. The filter member is composed of a large number of intersecting porous walls which define a large number of inlet gas passages and outlet gas passages which are adjacent to each other. The electric heater is composed of at least one film-shaped heating resistor which is directly formed on the upstream end surface of the filter member so as to be integral therewith. When the amount of carbon particulates collected by the filter member reaches a predetermined level, electric current is supplied to the electric heater. The carbon particulates adhered to the upstream end surface of the filter member are ignited and burnt off. Then, the combustion of carbon particulates spreads to the other carbon particulates collected in the other portion of the filter member.~~

[0012] ~~U.S. Pat. No. 5,843,304 teaches a materials treatment system which includes filtration and treatment of solid and liquid components of a material, such as a waste material. A filter or substrate assembly is provided which allows liquids to pass therethrough, while retaining solids. The solids are then incinerated utilizing microwave energy, and the liquids can be treated after passing through the filter element, for example, utilizing a treatment liquid such as an oxidant liquid. The filter assembly can also include an exhaust filter assembly which removes solids or particulate matter from exhaust gasses, with the retained solids/particulates incinerated utilizing microwave energy.~~

[0013] ~~U.S. Pat. No. 5,074,112 teaches a filter assembly for an internal combustion engine which includes, in combination, a housing defining an exhaust gas passage having an inlet end and an outlet end and a cavity intermediate the inlet and outlet ends thereof and in serial fluid communication therewith, the cavity defining an electromagnetically resonant coaxial line waveguide, a filter disposed within the cavity for removing particulate products of combustion from exhaust gases passing through the cavity, and a mechanism for producing axisymmetrically distributed, standing electromagnetic waves within the cavity whereby to couple electromagnetic energy in the waves into lossy material in the cavity to produce heat for incinerating the particulate products of combustion accumulated on the filter.~~

[0014] U.S. Pat. No. 4,934,141 teaches a device for microwave elimination of particles contained in the exhaust gases of diesel engines in which a microwave source and a conductor of the electromagnetic field generated by the source is joined with a resonator mounted on an element of the pipe for the exhaust gases which contains an insert, characterized by the fact that the insert consists of a filter whose upstream and downstream ends are offset toward the inside of the cavity defined by the resonator and delimit two chambers in which conductors of the electromagnetic field come out, respectively.

[0015] U.S. Pat. No. 4,825,651 teaches a device and method for separating soot or other impurities from the exhaust gases of an internal combustion engine, particularly a diesel internal combustion engine, comprises a microwave source that is coupled to the intermediate section of the exhaust pipe that is constructed for the development of an electromagnetic field, an effective burning of the soot with a low flow resistance, the intermediate section being developed as a cavity resonator and at its exhaust gas inlet and exhaust gas outlet, is equipped with a metal grid, and an insert made of a dielectric material in the cavity resonator concentrates the exhaust gas flow in the area of high energy density of the electromagnetic field.

[0016] U.S. Pat. No. 4,477,771 teaches conductive particulates in the form of soot which are collected from diesel engine exhaust gases on a porous wall monolithic ceramic filter in such a way that the soot is somewhat uniformly distributed throughout the filter. The filter is housed in a chamber having a property of a microwave resonant cavity and the cavity is excited with microwave energy. As the particulates are collected the cavity appears to the microwaves to have an increasing dielectric constant even though the matter being accumulated is conductive rather than dielectric so that as collected on the porous filter it has the property of an artificial dielectric. The response of the cavity to the microwave energy is monitored to sense the effect of the dielectric constant of the material within the cavity to provide a measure of the soot content in the filter.

[0017] U.S. Pat. No. 5,902,514 teaches a material for microwave band devices which are used by the general people and in industrial electronic apparatuses. Particularly, a magnetic ceramic composition for use in microwave devices, a magnetic ceramics for use in microwave devices and a preparation method therefore are disclosed, in which the saturation magnetization can be easily controlled, and a low ferri-magnetic resonance half line width and an acceptable curie temperature are ensured. The magnetic ceramic composition for microwave devices includes yttrium oxide (Y_2O_3), iron oxide (Fe_2O_3), tin oxide (SnO_2), aluminum oxide (Al_2O_3) and a calcium supply source. The magnetic ceramics for the microwave devices are manufactured by carrying out a forming and a sintering after mixing: yttrium oxide, iron oxide, tin oxide, aluminum oxide and calcium carbonate (or calcium oxide) based on a formula shown below. It has a saturation magnetization of 100-1,800 G at the normal temperature, a temperature coefficient for the saturation magnetization of 0.2%/degree. C., and a ferri-magnetic resonance half line width of less than 60 Oe, $Y(3-x)Ca(x/2)Sn(x/2)Fe(5-y)Al(y)O(12)$ where $0.1 \leq x \leq 1$, and $0.1 \leq y \leq 1.5$.

[0018] U.S. Pat. No. 5,843,860 teaches a ceramic composition for high frequency dielectrics which includes the main ingredients of ZrO_2 , SnO_2 and TiO_2 and a subsidiary ingredient of $(Mn(NO_3)_2 \cdot 4H_2O)$. A homogeneous ceramic composition can be prepared by a process which comprises the steps of: adding ZrO_2 , SnO_2 and TiO_2 by the molar ratio to satisfy $(ZrO_2)(1-x)(SnO_2)(x)(TiO_2)(1+y)$ (wherein, 0.1M .degree. C. or above; and, adding 1% or less of $Mn(NO_3)_2 \cdot 4H_2O$ by weight of MnO to the mixture. The ceramic composition of the invention has a high dielectric constant of 40 or more, a quality factor of 7000 or more, and a temperature coefficient of resonance frequency below 10. Accordingly, it can be used for an integrated circuit at microwave as well as at high frequency, or for dielectric resonators.

[0019] U.S. Pat. No. 5,808,282 teaches a microwave susceptor bed which is useful for sintering ceramics, ceramic composites and metal powders. The microwave susceptor bed contains granules of a major amount of a microwave susceptor material, and a minor

amount of a refractory parting agent, either dispersed in the susceptor material, or as a coating on the susceptor material. Alumina is the preferred susceptor material. Carbon is the most preferred parting agent. A sintering process uses the bed to produce novel silicon nitride products.

[0020] U.S. Pat. No. 5,446,270 teaches a composition which includes susceptors having the capability of absorbing microwave energy and a matrix. The susceptors includes a particulate substrate substantially non-reflective of microwave energy and a coating capable of absorbing microwave energy. The matrix is substantially non-reflective of microwave energy. Susceptors are typically particles having a thin film coating thereon. The matrix typically includes polymeric or ceramic materials that are stable at temperatures conventionally used in microwave cooking. The composition allows reuse of the susceptors, eliminates decline in heating rate, eliminates arcing, allows the heating rate to be controlled, allows overheating to be controlled, and allows formation of microwave heatable composite materials having very low metal content.

[0021] U.S. Pat. No. 5,365,042 teaches a heat treatment installation for parts made of a composite material which has a ceramic matrix and which includes a treatment enclosure. The treatment enclosure is connected to a microwave generator by a wave guide and which includes a press for hot pressing a part to be treated in the enclosure and a gas source for introducing a protective gas into the enclosure.

[0022] U.S. Pat. No. 5,126,529 teaches a method for forming a three dimensional object by thermal spraying which utilizes a plurality of masks positioned and removed over a work surface in accordance with a predetermined sequence. The masks correspond to cross sections normal to a centerline through the work piece. One set of masks defines all cross sections through the work piece. A second set of masks contains at least one mask. The mask corresponds to each mask of the first set. Masks from each set are alternatively placed above a work surface and sprayed with either a deposition material from which the work piece will be made or a complementary material. In this manner, layers of material form a block of deposition material and complementary

material. The complementary material serves as a support structure during forming and is removed. Preferably, the complementary material has a lower melting temperature than the deposition material and is removed by heating the block. Alternatively, one could mask only for the deposition material and remove complementary material overlying the deposition material after each spraying of complementary material.

[0023] U.S. Pat. No. 4,199,387 teaches an air filter unit of the pleated media, high efficiency type. The media pleat edges are sealed to the supporting frame to prevent bypass of air with a ceramic adhesive and fibrous ceramic mat which allows the unit to be exposed to high temperatures (e.g., up to 2000 degree. F.) without danger of seal breakdown. While in the form of a slurry the adhesive is applied, for example, with a trowel to the zig-zag pleated edges of the media which, together with corrugated spacers, forms the filter core. The latter is then surrounded on four sides by the compressible mat of fibrous ceramic material and inserted in a box-like support frame with the slurry filling the space between the pleated edges of the media and the fibrous mat. The filter core and the surrounding mat are assembled with the support frame while the slurry is still wet whereby, upon hardening, the resulting layers of ceramic cement provide a complete, heat-resistant seal while avoiding cracking in normal handling due to the resilience of the compressed fibrous mat which maintains an airtight seal between hardened ceramic and support frame.

[0024] U.S. Pat. No. 6,063,150 teaches a self cleaning particle filter for Diesel engines which includes a filter housing, control circuitry, a removable filter sandwich and independent power source. The removable filter sandwich includes a number of sintered metal strips sewn and positioned between two sheets of inorganic material to provide a filter sandwich. Current is delivered to the metal filter strips to efficiently burn off carbon, lube oil and unburned fuel particulates which have been filtered from exhaust gas. The filter sandwich is formed into a cylindrical configuration and mounted onto a perforated metal carrier tube for receiving and filtering exhaust gas.

[0025] U.S. Pat. No. 6,101,793 teaches an exhaust gas filter having a ceramic filter body is configured such that a specific heat h (cal/g.degree. C.) of ceramic powder constituting the body, and a bulk specific gravity d (g/cm³) of the filter, satisfy the relation $0.12 \text{ (cal/cm}^3\text{.degree. C.)} \leq h \cdot d \leq 0.19 \text{ (cal/cm}^3\text{.degree.C.)}$. The ceramic filter body includes a plurality of cells which extend axially to open at opposite ends of the body. One of the opposite axial ends of each of the cells is closed by a filler in such a manner that the closed ends of the cells and the open ends of the cells are arranged in an alternating configuration. The filter traps particulates in the exhaust gas, and the trapped particulates are removed by regeneration combustion of the filter. The filter exhibits excellent durability, thus preventing the formation of cracks in the surface and interior of the filter. When the filter is mounted on a diesel engine, the diesel engine advantageously does not discharge black smoke.

[0026] U.S. Pat. No. 5,756,412 teaches a dielectric ceramic composition for microwave applications which consists essentially of the compound having a formula $B'B(2)'O(6)$, wherein B' is at least one metal selected from the group of Mg, Ca, Co, Mn, Ni and Zn, and wherein B'' is one of Nb or Ta, and additionally includes at least one compound selected from the group of CuO, V₂O₅, La₂O₃, Sb₂O₅, WO₃, MnCO₃, MgO, SrCO₃, ZNO, and Bi₂O₃ as an additive, wherein the amount of the additive is 0.05% to 2.0% by weight of the total weight of the composition.

[0027] The synthesis of polycarbosilane from the pyrolytic condensation reaction of polydimethylsilane obtained from the reaction of dichlorodimethylsilane with an alkali metal, such as sodium. In the latter approach, polydimethylsilane can be prepared by Wurtz type coupling of dichlorodimethylsilane with sodium in toluene. The direct pyrolysis of polydimethylsilane, a viscous thermoplastic resin, at high temperature gives SiC in a ceramic yield of about 30% - 40%. By thermally cross linking the polydimethylsilane into an infusible rigid thermoset polymer, which is insoluble in any common solvents, the subsequent pyrolysis yield is on the order of 88% - 93%. This

~~thermolysis was accomplished by refluxing the polydimethyl silane to in excess of 350.degree. C.~~

[0028] ~~Numerous pre-ceramic polymers with improved yields of the ceramic have been described in U.S. Pat. No. 5,138,080, U.S. Pat. No. 5,091,271, U.S. Pat. No. 5,051,215 and U.S. Pat. No. 5,707,471. The fundamental chemistry contained in these embodiments is specific to the process employed and mainly leaves the pre-ceramic polymer in a thermoplastic state. These pre-ceramic polymers which catalytic or photo-induced cross linking do not satisfy the high ceramic yield, purity and fluidity in combination with low temperature crosslinking ability necessary for producing large densified ceramic structures in a single step continuous process.~~

[0029] ~~U.S. Pat. No. 5,138,080 teaches a novel polysilamethylenesilane polymers which has polysilane polycarbosilane skeleton which can be prepared in one step reaction from mixtures of chlorosilaalkanes and organochloro silanes with alkali metals in one of appropriate solvents or in combination of solvents thereof. Such polysilamethyleno silane polymers are soluble and thermoplastic and can be pyrolyzed to obtain improved yields of silicon carbide at atmospheric pressure.~~

[0030] ~~U.S. Pat. No. 5,091,271 teaches a shaped silicon carbide based ceramic article which has a mechanical strength which is produced at a high efficiency by a process including the step of forming an organic silicone polymer, for example, polycarbosilastyrene copolymer, into a predetermined shape, for example, a filament or film; doping the shaped polymer with a doping material consisting of at least one type of halogen, for example, bromine or iodine, in an amount of 0.01% to 150% based on the weight of the shaped polymer, to render the shaped polymer infusible; and pyrolyzing the infusible shaped polymer into a shaped SiC based ceramic article at a temperature of 800.degree. C. to 1400.degree. C. in an inert gas atmosphere, optionally the halogen-doped shaped polymer being treated with a basic material, for example, ammonia, before the pyrolyzing step, to make the filament uniformly infusible.~~

[0031] U.S. Pat. No. 5,300,605 teaches poly(I hydro-1-R-1-silapent-3-ene) homopolymers and copolymers which contain silane segments with reactive silicon hydride bonds and contain hydrocarbon segments with cis and trans carbon-carbon double bonds.

[0032] U.S. Pat. No. 5,171,810 teaches random or block copolymers with (I hydro-1-R-1-sila-cis-pent-3-ene), poly(I hydro-1-R-3,4-benzo-1-sila-pent-3-ene) and disubstituted 1-silapent-3-ene repeating units of the general formula ##STR1## where R is hydrogen, an alkyl radical containing from one to four carbon atoms or phenyl, R.¹ is hydrogen, an alkyl radical containing from one to four carbon atoms, phenyl or a halogen and R.² is hydrogen, or R.¹ and R.² are combined to form a phenyl ring, are prepared by the anionic ring-opening polymerization of silacyclopent-3-enes or 2-silaindans with an organometallic base and cation coordinating ligand catalyst system or a metathesis ring-opening catalyst system.

[0033] U.S. Pat. No. 5,169,916 Poly(I hydro-1-R-1-sila-cis-pent-3-ene) and poly(I hydro-1-R-3,4-benzo-1-sila-pent-3-ene) polymers which has repeating units of the general formula polycarbosilane containing at least two tbd.SiH groups per molecule via intimately contacting such fusible polycarbosilane with an effective hardening amount of the vapors of sulfur.

[0034] U.S. Pat. No. 5,064,915 teaches insoluble polycarbosilanes, readily pyrolyzed into silicon carbide ceramic materials such as SiC fibers, are produced by hardening a fusible polycarbosilane containing at least two tbd.SiH groups per molecule via intimately contacting such fusible polycarbosilane with an effective hardening amount of the vapors of sulfur.

[0035] U.S. Pat. No. 5,049,529 teaches carbon nitride ceramic materials which are produced by hardening a fusible polycarbosilane containing at least two tbd.SiH groups per molecule by intimately contacting such fusible polycarbosilane with an effective hardening amount of the vapors of sulfur, next, heat treating the infusible polycarbosilane

which results under an ammonia atmosphere to such extent as to introduce nitrogen into the infusible polycarbosilane without completely removing the carbon therefrom and then heat treating the nitrogenated polycarbosilane in a vacuum or in an inert atmosphere to such extent as to essentially completely convert it into a ceramic silicon carbon nitride.

[0036] U.S. Pat. No. 5,051,215 teaches a rapid method of infusibilizing pre-ceramic polymers which includes treatment of the polymers with gaseous nitrogen dioxide. The infusibilized polymers may be pyrolyzed to temperatures in excess of about 800 degree C. to yield ceramic materials with low oxygen content and, thus, good thermal stability. The methods are especially useful for the production of ceramic fibers and, more specifically, to the on-line production of ceramic fibers.

[0037] U.S. Pat. No. 5,028,571 teaches silicon nitride ceramic materials which are produced by hardening a fusible polycarbosilane containing at least two dbd.SiH groups per molecule by intimately contacting such fusible polycarbosilane with an effective hardening amount of the vapors of sulfur and then pyrolyzing the infusible polycarbosilane which results under an ammonia atmosphere.

[0038] U.S. Pat. No. 4,847,027 teaches a method for the preparation of ceramic materials or articles by the pyrolysis of pre-ceramic polymers wherein the pre-ceramic polymers are rendered infusible prior to pyrolysis by exposure to gaseous nitric oxide. Ceramic materials with low oxygen content, excellent physical properties, and good thermal stability can be obtained by the practice of this process. This method is especially suited for the preparation of ceramic fibers.

[0039] U.S. Pat. No. 5,714,025 teaches a method for preparing a ceramic forming pre-preg tape which includes the steps of dispersing in water a ceramic forming powder and a fiber, flocculating the dispersion by adding a cationic wet strength resin and an anionic polymer, dewatering the flocculated dispersion to form a sheet, wet pressing and drying the sheet, and coating or impregnating the sheet with an adhesive selected from the group consisting of a polymeric ceramic precursor, and a dispersion of an organic binder and

the materials used to form the sheet. The tape can be used to form laminates, which are fired to consolidate the tapes to a ceramic.

[0040] U.S. Pat. No. 5,707,471 teaches a method for preparing fiber reinforced ceramic matrix composites which includes the steps of coating refractory fibers, forming the coated fibers into the desired curing the coated fibers to form a pre-preg, heating the pre-preg to form a composite and heating the composite in an oxidizing shape, environment to form an in situ sealant oxide coating on the composite. The refractory fibers have a interfacial coating thereon with a curable pre-ceramic polymer which has a char containing greater than about 50% sealant oxide atoms. The resultant composites have good oxidation resistance at high temperature as well as good strength and toughness.

[0041] U.S. Pat. No. 5,512,351 teaches a new pre-preg material which has good tack drape properties and feasible out time. The pre-preg material is prepared by impregnating inorganic fibers with a composition which includes a fine powder of a metal oxide or oxides having an average particle diameter of not larger than one micrometer, a soluble siloxane polymer having double chain structure, a trifunctional silane compound having at least one ethylenically unsaturated double bond in the molecule thereof, a organic peroxide and a radically polymerizable monomer having at least two ethylenically unsaturated double bonds and heating the impregnated fibers.

[0042] U.S. Pat. No. 4,835,238 teaches a reaction of 1,1-dichloro-silacyclobutanes with nitrogen-containing difunctional nucleophiles which gives polysilacyclobutasilazanes which can be crosslinked and also converted to ceramic materials.

[0043] Numerous processing mechanics with various direct applications have been described, for example, in the U.S. Pat. No. 5,820,483, U.S. Pat. No. 5,626,707, U.S. Pat. No. 5,732,743 and U.S. Pat. No. 5,698,055. The process mechanics are for a single product process and do not permit continuous curing and pyrolysis in a single step to produce highly dense thick ceramic components.

[0044] U.S. Pat. No. 5,820,483 teaches methods for manufacturing a shaft for a golf club. A plug is detachably affixed to a distal end of a mandrel. A plurality of plies of pre-preg composite sheet are wrapped around the mandrel and plug and, thereafter, heated causing the resin comprising the various plies to be cured. The mandrel is then removed from the formed shaft, leaving the plug as an integral part of the distal tip of the shaft.

[0045] U.S. Pat. No. 5,626,707 teaches an apparatus which produces a composite tubular article. The apparatus includes a frame, a drive mechanism for rotating a mandrel, at least two spindles mounted to the frame, a tensioner and a belt extending between the first and second spindles. The apparatus may be used to roll pre-preg strips or similar sheets of composite materials around the mandrel. The belt travels over the spindles, and the spindles guide the belt through changes in its direction of travel. The mandrel is mounted in the drive mechanism in contact with the belt, which changes its direction of travel around the mandrel. The lower surface of the belt bears against upper portions of the spindles, and the mandrel contacts the upper surface of the belt. As the drive mechanism rotates the mandrel, pre-preg sheets are fed between the mandrel and the belt and are thereby wrapped around the mandrel. The belt presses the pre-preg sheets against the mandrel. The wrapped mandrel may then be removed from the apparatus and cured in any suitable manner known in the art to produce the a composite tubular article.

[0046] U.S. Pat. No. 5,732,743 teaches a method for joining and repairing pipes includes the step of utilizing photo-curable resins in the form of a fabric patch to for quickly repairing or sealing pipes. A photo-curable flexible pre-preg fabric is wrapped over the entire area of the pipe to be joined or repaired. The pre-preg fabric contains multiple layers of varying widths and lengths. The pre-preg fabric is then exposed to photo-radiation which cures and seals the pipe.

[0047] U.S. Pat. No. 5,698,055 teaches a method for making a reinforced tubular laminate. A dry braided fiber sleeve is placed between a mandrel and spiral tape wrap

either over, under, or layered with a pre-preg material. During the initial stages of the curing process, while the temperature is rising, the resin in the pre-preg material flows and wets out the dry braid. When the final cure takes place, the braid becomes an integral part of the finished laminate. The choice of fiber materials and braid angle permits various tubular laminate strengths. The selection of fiber colors and patterns permit a wide variety of tubular laminate aesthetic characteristics.

[0048] U.S. Pat. No. 5,632,834 teaches sandwich structures which are made of fiber-reinforced ceramics. The base substance of the ceramic matrix consists of a Si-organic polymer and a ceramic or metallic powder. A cross-linking of the Si-organic polymer takes place under increased pressure and at an increased temperature. After the joining of the facings and the honeycomb core, the sandwich structure is pyrolysed to form a ceramic material.

[0049] U.S. Pat. No. 5,641,817 teaches organometallic ceramic precursor binders which are used to fabricate shaped bodies by different techniques. Exemplary shape making techniques which utilize hardenable, liquid, organometallic, ceramic precursor binders include the fabrication of negatives of parts to be made (e.g., sand molds and sand cores for metalcasting, etc.), as well as utilizing ceramic precursor binders to make shapes directly (e.g., brake shoes, brake pads, clutch parts, grinding wheels, polymer concrete, refractory patches and liners, etc.). A thermosettable, liquid ceramic precursors provides suitable strength sand molds and sand cores at very low binder levels and, upon exposure to molten metal casting exhibit low emissions toxicity as a result of their high char yields of ceramic upon exposure to heat. The process involves the fabrication of preforms used in the formation of composite articles. Production costs, and relatively poor physical properties prohibits their inherently large cost of capitalization, high wide use.

[0050] U.S. Pat. No. 4,631,179 teaches this ring-opening polymerization reactions method to obtain a linear polymer of the formula $[\text{SiH}_{\text{sub.}2}\text{CH}_{\text{sub.}2}]_{\text{sub.}n}$. This polymer exhibit ceramics yields up to 85% on pyrolysis. The starting material for the ring-opening polymerization reaction was the cyclic compound

~~[SiH.sub.2CH.sub.2].sub.2, which is difficult and costly to obtain in pure form by either of the procedures that have been reported.~~

[0051] U.S. Pat. No. 5,888,641 teaches an exhaust manifold for an engine which are made of all fiber reinforced ceramic matrix composite material so as to be light weight and high temperature resistant. A method of making the exhaust manifold includes the steps of forming a liner of a cast monolithic ceramic material containing pores, filling the pores of the cast monolithic ceramic material with a pre-ceramic polymer resin, coating reinforcing fibers with an interface material to prevent a pre-ceramic polymer resin from adhering strongly to the reinforcing fibers, forming a mixture of a pre-ceramic polymer resin and reinforcing fibers coated with the interface material, forming an exhaust manifold-shaped structure from the mixture of the pre-ceramic polymer resin and the reinforcing fibers coated with the interface material by placing the mixture on at least a portion of the cast monolithic ceramic material, and firing the exhaust component-shaped structure at a temperature and a time sufficient to convert the pre-ceramic polymer resin to a ceramic thereby forming a reinforced ceramic composite.

Silicon carbide (SiC) is one of several advanced ceramic materials which are currently receiving considerable attention as electronic materials, as potential replacements for metals in engines, and for a variety of other applications where high strength, combined with low density and resistance to oxidation, corrosion and thermal degradation at temperatures in excess of 1000 °C are required. Unfortunately, these extremely hard, non-melting ceramics are difficult to process by conventional forming, machining, or spinning applications rendering their use for many of these potential applications problematic. In particular, the production of thin films by solution casting, continuous fiber by solution or melt spinning, a SiC matrix composite by liquid phase infiltration, or a monolithic object using a precursor-based binder/sintering aid, all require a source of SiC which is suitable for solution or melt processing and which possesses certain requisite physical and chemical properties which are generally characteristic of polymeric materials.

Polymeric precursors to ceramics such as SiC afford a potential solution to this problem as they would allow the use of conventional processing operations prior to conversion to ceramic. A ceramic precursor should be soluble in organic solvents, moldable or spinnable, crosslinkable, and give pure ceramic product in high yield on pyrolysis. Unfortunately, it is difficult to achieve all these goals simultaneously. Currently available SiC precursor systems are lacking in one or more of these areas. Problems have been encountered in efforts to employ the existing polysilane and polycarbosilane precursors to SiC for preparation of SiC fiber and monolithic ceramic objects. All of these precursors have C/Si ratios considerably greater than one, and undergo a complex series of ill-defined thermal decomposition reactions which generally lead to incorporation of excess carbon. The existence of even small amounts of carbon at the grain boundaries within SiC ceramics has been found to have a detrimental effect on the strength of the ceramic, contributing to the relatively low room-temperature tensile strengths typically observed for precursor-derived SiC fibers.

Efforts to develop polymeric precursors to SiC have focused largely on two types of polymers, polysilanes, which have a Si-Si backbone, and polycarbosilanes, in which the polymer backbone is [-Si--C--]_n. The polysilanes all suffer from problems due to insolubility, infusibility and/or excess carbon incorporation. Certain of the polycarbosilanes have more suitable physical properties for processing; however, in general, these also contain a higher-than-1:1 C:Si ratio and incorporate excess carbon on pyrolysis.

In the case of the polycarbosilanes, high molecular weight linear polymers of the type [R₂SiCH₂]_n, where R is H and/or hydrocarbon groups, have been prepared by ring-opening-polymerization reactions starting from cyclic disilacyclobutanes using chloroplatinic acid and related catalyst systems; however, such linear polycarbosilanes generally exhibit low yields of ceramic product on pyrolysis due to chain "unzipping" reactions. For example, studies of high molecular weight [Me₂SiCH₂]_n polymers have indicated virtually complete volatilization on pyrolysis under an inert atmosphere to 1000 °C.

Use of propargyl groups ($\text{HC}\equiv\text{CCH}_2$), such as propargyl chloride ($\text{HC}\equiv\text{CCH}_2\text{Cl}$), propargyl bromide ($\text{HC}\equiv\text{CCH}_2\text{Br}$), propargyl alcohol ($\text{HC}\equiv\text{CCH}_2\text{OH}$), propargyl magnesium chloride ($\text{HC}\equiv\text{CCH}_2\text{MgCl}$), propargyl calcium chloride ($\text{HC}\equiv\text{CCH}_2\text{CaCl}$), propargyl amine ($\text{HC}\equiv\text{CCH}_2\text{NH}_2$), and other propargyl-containing species introduces the photo-curable (cross-linkable) triple-bonded carbon linkages into the pre-ceramic polymer.

U.S. Pat. No. 5,153,295 teaches the use of preceramic polymers with an Si--C backbone structure, such as allylhydridopolycarbosilane (AHPSC), formed from the Grignard coupling reaction of a halomethylcarbosilane followed by reduction using a metal hydride in which either a UV cross-linkable ethynyl (i.e. acetylide) or propargyl group has been introduced into the polymer.

[0052] ~~U.S. Pat. No. 5,153,295 teaches compositions of matter which have potential utility as precursors to silicon carbide. These compositions are obtained by a Grignard coupling process starting from chlorocarbosilanes, a readily available class of compounds. The new precursors constitute a fundamentally new type of polycarbosilane that is characterized by a branched, $[\text{Si}-\text{C}]_{\text{n}}$ "backbone" which consists of $\text{SiR}_{\text{sub.3}}\text{CH}_{\text{sub.2}}$, $\text{SiR}_{\text{sub.2}}\text{CH}_{\text{sub.2}}$, $\text{.dbd.SiRCH}_{\text{sub.2}}$, and $\text{.tbd.SiCH}_{\text{sub.2}}$ units (where R is usually H but can also be other organic or inorganic groups, e.g., lower alkyl or alkenyl, as may be needed to promote crosslinking or to modify the physical properties of the polymer or the composition of the final ceramic product). A key feature of these polymers is that substantially all of the linkages between the Si--C units are "head-to-tail", i.e., they are Si to C. The polycarbosilane "SiH₂CH₂" has a carbon to silicon ratio of 1 to 1 and where substantially all of the substituents on the polymer backbone are hydrogen. This polymer consists largely of a combination of the four polymer "units": $\text{SiH}_{\text{sub.3}}\text{CH}_{\text{sub.2}}$, $\text{SiH}_{\text{sub.2}}\text{CH}_{\text{sub.2}}$, $\text{.dbd.SiHCH}_{\text{sub.2}}$, and $\text{.tbd.SiCH}_{\text{sub.2}}$ which are connected "head to tail" in such a manner that a complex, branched structure results. The branched sites introduced by the last two "units" are offset by a corresponding number of $\text{SiH}_{\text{sub.3}}\text{CH}_{\text{sub.2}}$ "end groups" while maintaining the alternating Si--C "backbone". The relative numbers of the polymer "units" are such~~

that the "average" formula is $\text{SiH}_{\text{sub.}2}\text{CH}_{\text{sub.}2}$. These polymers have the advantage that it is only necessary to lose hydrogen during pyrolysis, thus ceramic yields of over 90% are possible, in principle. The extensive Si--H functionality allows facile crosslinking and the 1 to 1 carbon to silicon ratio and avoids the incorporation of excess carbon in the SiC products that are ultimately formed. The synthetic procedure employed to make them allows facile modification of the polymer, such as by introduction of small amounts of pendant vinyl groups, prior to reduction. The resulting vinyl-substituted " SiH_2CH_2 " polymer has been found to have improved crosslinking properties and higher ceramic yield.

[0053] A pre-ceramic polymer is has been prepared by a thermally induced methylene insertion reaction of polydimethylsilane. The resulting polymer is only approximately represented by the formula $[\text{SiHMeCH}_2]_n$, as significant amounts of unreacted $(\text{SiMe}_2)_n$ units, complex rearrangements, and branching are observed. ~~Neither the preparation nor the resulting structure of this precursor are therefore similar to the instant process.~~ In addition to the carbosilane "units", large amounts of Si--Si bonding remains in the "backbone" of the polymer. This polymer, in contrast to the instant process, disadvantageously contains twice the stoichiometric amount of carbon for SiC formation. The excess carbon must be eliminated through pyrolytic processes that are by no means quantitative. Despite the shortcomings, this polymer has been employed to prepare "SiC" fiber. However, it must be treated with various crosslinking agents prior to pyrolysis which introduce contaminants. This results in a final ceramic product that contains significant amounts of excess carbon and silica which greatly degrade the high temperature performance of the fiber.

[0054] SiC precursors, predominately linear polycarbo-silanes, have been prepared via potassium dechlorination of chloro-chloromethyl-dimethylsilane. The resulting polymers have not been fully characterized, but probably contain significant numbers of Si--Si and $\text{CH}_2\text{--CH}_2$ groups in the polymer backbone. The alkali metal dechlorination process used in the synthesis of such materials does not exhibit the selective head-tail coupling found with Grignard coupling. The pendant methyl groups in such materials also lead to the

incorporation of excess carbon into the system. In several polymer systems mixtures containing vinylchlorosilanes (such as $\text{CH}_2\equiv\text{CH--Si}(\text{Me})\text{Cl}_2$) and Me_2SiCl_2 are coupled by dechlorination with potassium in tetrahydro-furan. U.S. Pat. No. 4,414,403 and U.S. Pat. No. 4,472,591 both teach this method. The "backbone" of the resulting polymers consists of a combination of Si--Si and Si-- $\text{CH}_2\text{CH}(\text{--Si})_2$ units. Later versions of this polymer $\text{Me}(\text{H})\text{SiCl}_2$ in addition to the Me_2SiCl_2 and are subjected to a sodium-hydrocarbon dechlorination process which does not attack vinyl groups. The resulting polymer consists of a predominately linear, Si--Si "backbone" bearing pendant methyl groups, with some Si--H and Si-- $\text{CH}\equiv\text{CH}_2$ functionality to allow crosslinking on pyrolysis.

[0055] None of these precursors derived using vinylchlorosilanes are similar to those of the process in that having predominantly Si--Si bonded "backbones", they are essentially polysilanes, not polycarbosilanes. In addition, the carbon in these polymers is primarily in the form of pendant methyl functionality and is present in considerable excess of the desirable 1 to 1 ratio with silicon. The ceramic products obtained from these polymers are known to contain considerable amounts of excess carbon.

[0056] Polymeric precursors to SiC have been obtained by redistribution reactions of methyl-chloro-disilane ($\text{Me}_{6-x}\text{Cl}_x\text{Si}_2$, $x=2-4$) mixtures, catalyzed by tetraalkyl-phosphonium halides which U.S. Pat. No. 4,310,481, U.S. Pat. No. 4,310,482 and U.S. Pat. No. 4,472,591 teach. In a typical preparation, elemental analysis of the polymer was employed to suggest the approximate formula $[\text{Si}(\text{Me})_{1.15}(\text{H})_{0.25}]_n$, with n averaging about 20. The ~~reaction is fundamentally different than that involved in the process and the~~ structures of the polymers ~~are also entirely different, involving involve~~ what is reported to be a complex arrangement of fused polysilane rings with methyl substitution and a polysilane backbone.

[0057] The formation of carbosilane polymers with pendent methyl groups has been prepared as by-products of the "reverse-Grignard" reaction of chloromethyl-dichloro-methylsilane. The chief purpose of this work was the preparation of carbosilane rings and

the polymeric byproduct was not characterized in detail nor was its use as a SiC precursor suggested. Studies of this material indicate that it has an unacceptably low ceramic yield on pyrolysis. These polymers are related to those described in the instant process and are obtained by a similar procedure, however, they contain twice the required amount carbon necessary for stoichiometric silicon carbide and their use as SiC precursors was not suggested. Moreover, the starting material, chloromethyl-dichloro-methylsilane, contains only two sites on the Si atom for chain growth and therefore cannot yield a structure which contains ~~tbd.SiCH₂~~ chain units. On this basis, the structure of the polymer obtained, as well as its physical properties and pyrolysis characteristics, is not optimal for use as an SiC precursor. must be significantly different from that of the subject process.

[0058] U.S. Pat. No. 4,631,179 teaches a polymer which is a product of the ring-opening polymerization of $(\text{SiH}_2\text{CH}_2)_2$ also has the nominal composition " SiH_2CH_2 ". However, the actual structure of this polymer is fundamentally and functionally different from that of the instant process. Instead of a highly branched structure comprised of $\text{SiR}_{\text{sub.3}}\text{CH}_{\text{sub.2}}$, $\text{SiR}_{\text{sub.2}}\text{CH}_{\text{sub.2}}$, $\text{dbd.SiRCH}_{\text{sub.2}}$, and $\text{tbd.SiCH}_{\text{sub.2}}$ units, the Smith polymer is reported to be a linear polycarbosilane which presumably has only $[\text{SiH}_2\text{CH}_2]$ as the internal chain segments. Such a fundamental structural difference would be expected to lead to quite different physical and chemical properties. The fundamental difference in these two structures has been verified by the preparation of a linear polymer analogous to polymer and the comparison of its infrared and H-NMR spectra.

[0059] Another important difference between the process of Smith and the instant process is the method used to obtain the product polymer and the nature of the starting materials. The $(\text{SiH}_2\text{CH}_2)_2$ monomer used by Smith is difficult and expensive to prepare and not generally available, whereas the chlorocarbosilanes used in the instant process are readily available through commercial sources.

[0060] U.S. Pat. No. 4,923,716 teaches chemical vapor deposition of silicon carbide which uses a "single molecular species" and which provides reactive fragments

containing both silicon and carbon atoms in equal number this process. Linear and cyclic structures of up to six units are mentioned. These compounds, which include both silanes and carbosilanes, are specifically chosen to be volatile for chemical vapor deposition use, and are distinctly different from the instant process, where the products are polymers of sufficiently high molecular weight that they cross-link before significant volatilization occurs. Such volatility would be highly undesirable for the applications under consideration for the polymers of the instant process, where excessive loss of the silicon-containing compound by vaporization on heating would be unacceptable.

[0061] The inventors hereby incorporate the above referenced patents and articles into this application.

SUMMARY OF THE INVENTION

[0062] The present invention is generally directed to a process of forming a photo-curable pre-ceramic polymer for use in for fabricating ceramic matrix composites.

[0063] In a first separate aspect of the invention the process includes the steps to silicon carbide ceramic.

[0064] In a second separate aspect of the invention the process includes the steps of reacting sodium acetylide with organo-chlorosilanes and condensing (polymerizing) the resultant organo-(ethynyl)-chloro silane product of step a with an excess of an alkali metal. The process includes the steps preparing a solution of thermoplastic photo-curable pre-ceramic polymer, passing a fiber, tape or fabric through the solution of thermoplastic photo-curable pre-ceramic polymer, applying the pre-preg to a shaped mandrel, using light energy to induce cross-linking of the photo-curable pre-ceramic polymer after application to the mandrel. The thermoplastic pre-ceramic polymer is cured and pyrolyzing the cured thermoplastic pre-ceramic polymer matrix composite material can be pyrolyzed to prepare a ceramic material.

[0065] In a third separate aspect of the invention a single-step process for fabricating continuous ceramic fiber ceramic matrix composites employs a thermoplastic photo-curable pre-ceramic polymer in which the component is shaped by a variety of standard composite fabrication techniques, such as filament winding, tape winding, and woven cloth winding. The process includes steps of passing ceramic fiber monofilament, tow, mat, or woven cloth through a solution of the thermoplastic photo-curable pre-ceramic polymer, applying ceramic fiber monofilament, tow, mat, or woven cloth to a shaped mandrel, using photo-energy of the ultraviolet, visible or infrared light spectrum to induce cross-linking (curing) of the photo-curable pre-ceramic polymer after application to the mandrel, and either partially or completely pyrolyzing the now cured pre-ceramic polymer matrix composite material.

[0066] Other aspects and many of the attendant advantages will be more readily appreciated as the same becomes better understood by reference to the following detailed description.

[0067] The features of the present invention which are believed to be novel are set forth with particularity in the appended claims.

DESCRIPTION OF DRAWINGS

[0068] FIG. 1 is a schematic drawing of an apparatus for making flat plates of ceramic composites from photo-curable pre-ceramic polymers.

[0069] FIG. 2 is a schematic drawing of an apparatus for making cylinders of ceramic composites from photo-curable pre-ceramic polymers.

DESCRIPTION OF THE PREFERRED EMBODIMENT

[0070] A continuous single step manufacturing process for fabricates dense low-porosity composites using novel cross-linkable pre-ceramic polymers and simple plastic industry technology adapted to the thermoset capability of the pre-ceramic polymer. The process

eliminates the multi-cycle polymer impregnation pyrolysis method. The process is a simple controllable production process for fiber reinforced ceramic matrix composites, which can be easily automated into large manufacturing continuous processes. This process combines high-yield cross-linkable pre-ceramic polymers and a single step automated process mechanism to produce ceramic components on the scale of aircraft fuselages, boat hulls, and large single ceramic sheets for space vehicle skin panels. The process provides chemically modified pre-ceramic polymers which are very fluid at temperatures 60 °C-100 °C, have high ceramic yields by weight of 75-95%, exhibit high purity and can be crosslinked into a thermoset with ultraviolet radiation.

The process achieves synthesizes these polymers by a series of chemical substitutions using commercially available polymers to incorporate ethynyl side groups on the polymers, which then. The resulting polymers contain unstable carbon triple bonds and ~~cross-link~~, are cross-linked by hydrosilylation with Si--H groups upon photo-exposure. The process ~~is to use the~~ uses chemical substitution ethynyl side group chemistry to produce SiC, Si₃N₄, Al₂O₃ and Al₃N₄ and TiC upon pyrolysis after photo-exposure. Conversion of precursor polymers like polycarbosilane and polysilazane to poly(ethynyl)carbosilane and poly(ethynyl)silazane achieve this objective.

In an embodiment of the invention, The process draws a fiber, tape, fabric, or woven cloth is drawn onto a mandrel or suitable substrate, by first passing it through the chemically modified pre-ceramic polymer. The objective of this process is to saturate the fiber, tape, fabric, woven cloth with the very fluid pre-ceramic polymer and then photo-cure it on the mandrel or substrate as the saturated material is drawn along by motorized winding or pulling mechanisms known to the prior art. The process provides a continuous fabrication process to enable making a dense (total porosity <8%) fiber reinforced ceramic composite in a single step. This objective is achieved by compacting each layer of pre-ceramic polymer saturated material onto the already pyrolyzed layer below permitting excess polymer to impregnate this layer. The back-fill allowed here reduces the final component porosity, increases strength and provides a short path for

volatiles to escape, mitigating ~~mitigating~~ interlayer delamination. This layer by layer buildup is continued until the required component thickness is reached.

[0071] The novel nature of the photocurable pre-ceramic polymer enables a process, which is unique to porous filters not achievable with conventional pre-ceramic polymers. This process employs the ability to thermoset the pre-ceramic polymer into a rubbery hard solid prior to heating. In this form the pre-ceramic polymer can be heated and subsequently pyrolyzed without flowing into unwanted areas of the filter. Because of the ability of this process to produce high yield beta-SiC in near Si--C stoichiometry a matrix or coating is formed upon sintering that is highly receptive to heating with microwave energy. The microwave susceptible porous filter is ideally suited for trapping particulate from diesel engine exhausts and can be regeneratively used by microwave heating to a temperature above the oxidation threshold of the trapped particulate soot. The pre ceramic polymer can be made to form not only SiC but also other ceramic bodies such as Si₃N₄, BC, LAS, etc.

[0072] Referring to FIG. 1, an apparatus 10 for making flat plates of ceramic composites from photo-curable pre-ceramic polymers includes a frame 11 with a process bed, a set of fabric rollers 12, a set of guide rollers 13, a set of drive rollers 14, a drive motor 15, a compression roller 16, a process head 17 having a light-emitting lamp, a furnace 18, a covering 19 and a source of inert gas and a control panel 20. The source of inert gas provides an inert atmosphere.

[0073] Referring to FIG. 2, an apparatus 110 for making ~~elyinders~~ cylinders of ceramic composites from photo-curable pre-ceramic polymers includes a dry nitrogen environmental chamber 111, a fabric roller 112, an applicator 113 of a photo-curable pre-ceramic polymer, a take-up mandrel 114, a pressure loaded compaction roller 115, a light-emitting lamp 116 and a consolidation and pyrolysis zone 117. The consolidation and pyrolysis zone 117 has a heater 118. The fabric roller dispenses woven ceramic fabric.

[0074] Commercially available polycarbosilanes and polycarbosiloxane polymers could be rendered photo-curable, by high intensity photo-radiation, through the addition of ethynyl side groups to the polymer. The polymer, poly(ethynyl) carbosilane, is rendered into an infusible thermoset upon photo-radiation. The process is able to similarly elevate ceramic yields to about 85% by weight.

[0075] It has been demonstrated that various combinations of di-functional and tri-functional silane precursors can be utilized to enhance cross-linking and elevate ceramic yield. Combinations of dichlorodimethylsilane (di-functional) and trichlorophenylsilane (tri-functional) can be employed. Through the addition of branching, or cross-linking, ceramic yields as high as 77% have been obtained. Further, it is possible to ~~doped~~ dope these polymers, with boron for example, to control sintering and crystallization behavior.

[0076] While this process allows the addition of ethynyl side groups to potentially a wide range of available pre-ceramic polymers, there are other methods of directly synthesizing poly(ethynyl)carbosilane, which are outlined below. In both of the following reaction paths, tri-functional organotrichlorosilanes are utilized, in part or in entirety, to permit the introduction of photo-polymerizable side-groups, such as ethynyl groups derived from the reaction of sodium acetylide with chlorosilane.

[0077] In the first reaction route, sodium acetylide is reacted with the organotrichlorosilane, such as a methyl- or phenyltrichlorosilane, as shown in step 1. Typically, this is performed in a solvent, such as hexane or methylene chloride. The by-product of this reaction is sodium chloride, which is insoluble and can be easily removed by filtration and/or sedimentation (step 2). The resulting organo(ethynyl)chlorosilane can be reacted directly with sodium which is a Wurtz type condensation reaction or mixed with an organodichlorosilane prior to the initiation of polycondensation. Assuming that all "R"s are the same, and "a+b=1", then the following reaction path can be proposed:

[0078] TABLE 1 New processing route 1: Steps and reaction chemistries to form poly (ethynyl) carbosilane--Final Product = $1/n \{SiR_{(a+2b)}C\equiv CH_{ag}\}_n$.

Processing Step	Reaction
1. The addition of a ethynyl (acetylide) side groups to tri-functional polysilazane reactant.	$a \{RSiCl_3 + g NaC\equiv CH \rightarrow RSiCl_{(3-g)}C\equiv CH_g + g NaCl\}$
2. Remove NaCl by filtration.	$-ag NaCl$
3. The addition of di-functional chain former (optional).	$+b \{R_2SiCl_2\}$
4. Condensation of modified precursor solution to produce poly (ethynyl) silazane pre-ceramic polymer through the addition of sodium.	$a (RSiCl_{(3-g)}C\equiv CH_g) + b (R_2SiCl_2) + 2[a(3-g) + b]Na \rightarrow 1/n \{Si_{(a+b)}R_{(a+2b)}C\equiv CH_{ag}\}_n + 2[a(3-g) + b]NaCl$

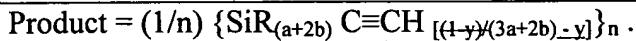
[0079] In route 1, the photo-cross-linkable ethynyl group (acetylide) is added prior to the initiation of Würtz type condensation reaction. In route 2, a method of adding ethynyl side-groups are added post-condensation, thereby avoiding the exposure of the ethynyl ligand to sodium during the pre-ceramic polymer synthesis is disclosed. In this process, tri-functional chlorosilanes, or and/or a mixture of di-functional and tri-functional chlorosilanes, are reacted with a sub-stoichiometric quantity of metallic sodium, sufficient to bring about an increase in molecular weight and viscosity of the now pre-ceramic polymer backbone, but leaving a fraction of the chlorosilane reaction sites unreacted. The resulting sodium chloride by-product can be removed by filtration and/or sedimentation (step 2).

[0080] Following polymer condensation, with unreacted chlorosilane sites intact, excess sodium acetylide is added to react with the aforementioned unreacted sites to produce poly(ethynyl)carbosilane photo-curable pre-ceramic polymer. The poly(ethynyl)carbosilane pre-ceramic polymer can be retrieved by solvent evaporation by

the application of heat and/or in vacuo. Assuming that all "R"s are the same, and "a+b=1", the final desired reaction product is expressed in the reaction path below in Table 2.

[0081] TABLE 2 New processing route 2: Steps, and reaction chemistries, to form poly(ethynyl)carbosilane.

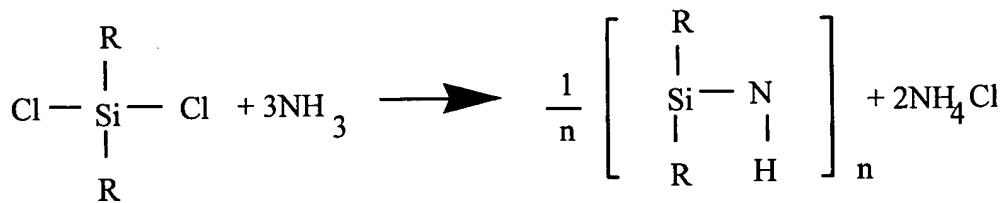
Processing Step	Reaction
1. Mixture of difunctional and trifunctional <u>chlorosilanes</u> <u>chlorosilanes</u> reacted with a sub-stoichiometric amount of sodium (where $y < [3a+2b]$).	$a(RsiCl_3) + b(R_2SiCl_2) + [y/(3a+2b)]Na \rightarrow$ $(1/n)\{Si_{(a+b)}R_{(a+2b)}Cl_{[(1-y)/(3a+2b)-y]}\}_n +$ $[y/(3a+2b)] NaCl$
2. Remove NaCl by filtration and/or sedimentation.	$-[y/(3a+2b)] NaCl$
3. Addition of ethynyl side groups to partially condensed polysilazane polymer through the addition of excess sodium acetylide.	$(1/n)\{Si_{(a+b)}R_{(a+2b)}Cl_{[(1-y)/(3a+2b)-y]}\}_n +$ $[(1-y)/(3a+2b)-y] NaC \equiv CH \rightarrow$ $(1/n)\{Si_{(a+b)}R_{(a+2b)}C \equiv CH_{[(1-y)/(3a+2b)-y]}\}_n +$ $[(1-y)/(3a+2b)-y] NaCl$



[0082] In the previous section, the method of preparing poly(ethynyl)carbosilane, a photo-curable pre-ceramic polymer precursor to silicon carbide has been reviewed. In this section, several of the promising methods of synthesizing polysilazane precursors to silicon nitride (Si_3N_4) and a method of conversion to poly(ethynyl)silazane, a photo-curable pre-ceramic polymer precursor to high yield Si_3N_4/SiC ceramic matrix composites are described. Si_3N_4 doped with 10-15 weight percent SiC has significantly lower creep at high temperature than pure Si_3N_4 . The creep rate at the minimum was lower by a factor of three than that of the unreinforced, monolithic matrix of equal grain

size. Thus, other researchers have recognized the potential importance of $\text{Si}_3\text{N}_4/\text{SiC}$ nanocomposite matrices for continuous ceramic fiber reinforced composites used in high temperature applications. Two advantages of the process of the process are the ability to fabricate large-scale composites employing existing polymer composite fabrication techniques due to the addition of the photo-cross-linkable ethynyl side-groups and the inclusion of the carbon containing ethynyl group should lead to the addition of approximately 5 to 20 weight percent SiC upon pyrolysis.

[0083] One of the simplest and direct methods of preparing polysilazane precursors to silicon nitride, with a 70 weight percent ceramic yield is to dissolve dichlorosilane in dichloromethane to yield polysilazane oils. Pyrolysis in flowing nitrogen gas yielded nearly phase pure a- Si_3N_4 after heat treatment at 1150 °C for 12 hours. Numerous other permutations and refinements to the preparation of polysilazane oils and polymers have been developed. The reaction path of polysilazane formation using dichlorosilanes and ammonia is set out below:



[0084] A number of the most direct permutations include the use of trichlorosilanes, methyltrichlorosilanes, dimethyldichlorosilanes, and vinyl-, butyl-, phenyl-, ethyl-, and hexyl-modified chlorosilanes. Increased molecular weight, and correspondingly increased ceramic yield, can be achieved by catalytically enhancing the cross-linking during final polymer preparation. A number of novel methods, including the use of ruthenium compounds and potassium hydride have been demonstrated to give ceramic yields upon pyrolysis as high as 85 percent. The inducement of cross-linking prior to pyrolysis is essential to achieving the high ceramic yields necessary to large-scale commercialization of Si_3N_4 matrix composites for high temperature applications. The

cross-linking methods cited in the literature, however, are chemical catalysts, making the shaping and forming processes extremely difficult.

[0085] A ceramic matrix of predominantly silicon nitride with about 10-15% SiC by weight is nearly ideal for fabricating CMCs. In addition, the catalytic cross-linking of the polysilazane precursor dramatically increases ceramic yield. The synthesis route should produce high yield $\text{Si}_3\text{N}_4/\text{SiC}$ nanocomposites employing a photocurable pre-ceramic polymer precursor.

[0086] One possible method would be to synthesize the unmodified polysilazane through the ammonolysis of various chlorosilane reactants in dichloromethane solvent followed by modifying the resulting polysilazanes, using a previously described process of chlorination followed by attachment of the ethynyl through reaction with sodium acetylide. Another alternative approach starts with a variety of dichlorosilanes and/or trichlorosilanes and reacts them with sodium acetylide at various concentrations, followed by ammonolysis to result in the final poly(ethynyl) silazane precursor as specifically detailed in the Table 3 below:

[0087] TABLE 3 Processing steps and reaction chemistries to form poly (ethynyl) silazane

Processing Step	Reaction
1. The addition of acetylide side groups <u>to</u> trifunctional polysilazane reactant.	$a \{ \text{RSiCl}_3 + g \text{ NaCCH} \rightarrow \text{RSiCl}_{(3-g)}\text{CCH}_g + g \text{ NaCl} \}$
2. Remove NaCl by filtration.	-ag NaCl
3. The addition of di-functional chain former	+b $\{ \text{R}_2\text{SiCl}_2 \}$
4. ammonolysis of modified precursor solution to produce poly(ethynyl)silazane pre-ceramic polymer	$a (\text{RSiCl}_{(3-g)}\text{CCH}_g) + b (\text{R}_2\text{SiCl}_2) + \text{NH}_3 \rightarrow b \{ [\text{SiR}_2(\text{NH})]_n \} + a \{ [\text{RSi}(\text{NH})_{(3-g)}\text{CCH}_g]_m \} + 2[a(3-g) + 2b]\text{NH}_4\text{CL}$

[0088] The following are examples of combining commercially available polymers and catalysts to achieve a final photo-curable pre-ceramic polymer to SiC ceramics. In order to be photo-curable, the polymer requires either double-bonded carbons such as Allyl side groups or triple-bonded carbons such as acetylide or propargyl side groups. The catalysts can include a thermally curable component such as benzoil peroxide and a photo-curable initiator such as Ciba-Geigy's Irgacure 1800™ or a combination of camphorquinone and 2-(dimethylamino)-ethyl methacrylate).

EXAMPLE 1

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Ciba-Geigy's Irgacure 1800	0.02
Photoinitiator 2	None	None

EXAMPLE 2

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Ciba-Geigy's Irgacure 1800	0.02
Photoinitiator 2	N,N-didihydroxyparatoluidine	0.02

EXAMPLE 3

Category	Compound	Amount (grams)

Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Ciba-Geigy's Irgacure 1800	0.01
Photoinitiator 2	None	None

EXAMPLE 4

Category	Compound	Amount (grams)
Polymer	Poly(ethynyl)carbosilane	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Ciba-Geigy's Irgacure 1800	0.02
Photoinitiator 2	None	None

EXAMPLE 5

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Camphorquinone	0.02
Photoinitiator 2	2-(dimethylamino)ethyl methacrylate	0.02

EXAMPLE 6

Category	Compound	Amount (grams)
Polymer	Poly(ethynyl)carbosilane	2.0

Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Camphorquinone	0.02
Photoinitiator 2	2-(dimethylamino)ethyl methacrylate	0.02

EXAMPLE 7

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	None
Photoinitiator 1	Camphorquinone	0.02
Photoinitiator 2	2-(dimethylamino)ethyl methacrylate	0.02

EXAMPLE 8

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	0.02
Photoinitiator 1	Camphorquinone	0.01
Photoinitiator 2	2-(dimethylamino)ethyl methacrylate	0.01

EXAMPLE 9

Category	Compound	Amount (grams)
Polymer	Allylhydridopolycarbosilane (5% allyl groups)	2.0
Catalyst	Benzoil Peroxide	None

Photoinitiator 1	Camphorquinone	0.01
Photoinitiator 2	2-(dimethylamino)ethyl methacrylate	0.01

[0096] All of the above examples cross-linked under photo-irradiation (using either ultraviolet light or blue light as indicated) within a few minutes to an hour under continuous irradiation at room temperature. The samples were transformed by this method from thermoplastic to thermoset pre-ceramic polymers which did not flow or deform significantly upon subsequent heat-treatment and pyrolysis, ultimately yielding SiC containing ceramics. The examples are meant to be illustrative. A person trained in the art can easily modify the ratios and selection of both pre-ceramic polymer and/or photo-initiators and catalyst combinations.

[0097] This process enables the continuous manufacture of fiber reinforced ceramic composites by the use of high ceramic yield pre-ceramic polymers which are photocurable to a thermoset from a thermoplastic state. A composite in any form or shape is fabricated by photocuring each individual layer of fiber with in-situ pyrolysis of the pre-ceramic polymer impregnated into the fiber layer. The layer by layer of fiber, fabric or woven cloth is pressure loaded to press the thermoplastic polymer infiltrated fabric onto the mandrel or flat substrate thereby permitting excess polymer to impregnate the porous, already pyrolyzed, layer below. This single step process allows a shorter mean free path for volatiles to escape with less destruction than the removal of organics from more massive parts, for consolidation of each layer individually, and for increased layer to layer bonding and improved interlaminar shear strengths.

[0098] ~~Silieon carbide (SiC) is one of several advanced ceramic materials which are currently receiving considerable attention as electronic materials, as potential replacements for metals in engines, and for a variety of other applications where high strength, combined with low density and resistance to oxidation, corrosion and thermal degradation at temperatures in excess of 1000 °C are required. Unfortunately, these~~

~~extremely hard, non melting ceramics are difficult to process by conventional forming, machining, or spinning applications rendering their use for many of these potential applications problematic. In particular, the production of thin films by solution casting, continuous fiber by solution or melt spinning, a SiC matrix composite by liquid phase infiltration, or a monolithic object using a precursor based binder/sintering aid, all require a source of SiC which is suitable for solution or melt processing and which possesses certain requisite physical and chemical properties which are generally characteristic of polymeric materials.~~

[0099] ~~Polymeric precursors to ceramics such as SiC afford a potential solution to this problem as they would allow the use of conventional processing operations prior to conversion to ceramic. A ceramic precursor should be soluble in organic solvents, moldable or spinnable, crosslinkable, and give pure ceramic product in high yield on pyrolysis. Unfortunately, it is difficult to achieve all these goals simultaneously. Currently available SiC precursor systems are lacking in one or more of these areas. Problems have been encountered in efforts to employ the existing polysilane and polycarbosilane precursors to SiC for preparation of SiC fiber and monolithic ceramic objects. All of these precursors have C/Si ratios considerably greater than one, and undergo a complex series of ill-defined thermal decomposition reactions which generally lead to incorporation of excess carbon. The existence of even small amounts of carbon at the grain boundaries within SiC ceramics has been found to have a detrimental effect on the strength of the ceramic, contributing to the relatively low room-temperature tensile strengths typically observed for precursor-derived SiC fibers.~~

[0100] ~~Efforts to develop polymeric precursors to SiC have focused largely on two types of polymers, polysilanes, which have a Si-Si backbone, and polycarbosilanes, in which the polymer backbone is [Si-C]._n The polysilanes all suffer from problems due to insolubility, infusibility and/or excess carbon incorporation. Certain of the polycarbosilanes have more suitable physical properties for processing; however, in general, these also contain a higher than 1:1 C:Si ratio and incorporate excess carbon on pyrolysis.~~

[0101] In the case of the polycarbosilanes, high molecular weight linear polymers of the type $[R_{sub.2}SiCH_{sub.2}]_{sub.n}$, where R is H and/or hydrocarbon groups, have been prepared by ring-opening polymerization reactions starting from cyclic disilacyclobutanes using chloroplatinic acid and related catalyst systems; however, such linear polycarbosilanes generally exhibit low yields of ceramic product on pyrolysis due to chain "unzipping" reactions. For example, studies of high molecular weight $[Me_{sub.2}SiCH_{sub.2}]_{sub.n}$ polymers have indicated virtually complete volatilization on pyrolysis under an inert atmosphere to 1000 degree C.

[0102] Use of propargyl groups ($HC_{sub.2}C=CH_2$), such as propargyl chloride ($HC_{sub.2}C=CH_2Cl$), propargyl bromide ($HC_{sub.2}C=CH_2Br$), propargyl alcohol ($HC_{sub.2}C=CH_2OH$), propargyl magnesium chloride ($HC_{sub.2}C=CH_2MgCl$), propargyl calcium chloride ($HC_{sub.2}C=CH_2CaCl$), propargyl amine ($HC_{sub.2}C=CH_2NH_2$), and other propargyl containing species introduces the photo-curable (Cross-linkable) triple bonded carbon linkages into the pre-ceramic polymer.

[0103] U.S. Pat. No. 5,153,295 teaches the use of ceramic polymers with an Si-C backbone structure, such as allylhydridopolycarbosilane (AHPCS), formed from the Grignard coupling reaction of a halomethylcarbosilane followed by reduction using a metal hydride in which either a UV cross-linkable ethynyl (i.e. acetylide) or propargyl group has been introduced into the polymer by methodologies described previously.

[0104] In general, The use of other ethynyl containing reagents, such as 1-ethynyl-1-cyclohexanol and 1,1'-ethynylene dicyclohexanol, can be directly coupled, due to the presence of hydroxyl (OH) bonds, to either halosilane (Si--X, where X.dbd.F, Cl, Br) and/or silanol (Si--OH) groups in the pre-ceramic polymer. The use of benzoyl peroxide or other chemical catalysts can be used in conjunction with double or triple bonded carbon side groups within the pre-ceramic polymer to achieve crosslinking.

[0106] A thermoplastic photo-curable pre-ceramic polymer as disclosed herein may be employed in a single-step fabrication process of continuous ceramic fiber ceramic matrix composites, employs a thermoplastic photo-curable pre-ceramic polymer in which the component is shaped by a variety of standard composite fabrication techniques, such as filament winding, tape winding, and woven cloth winding. The process includes For example, the process may include steps of passing ceramic fiber monofilament, tow, mat, or woven cloth through a solution of the thermoplastic photo-curable pre-ceramic polymer, applying ceramic fiber monofilament, tow, mat, or woven cloth to a moving flat substrate and using a heated or unheated compaction roller to press the thermoplastic pre-ceramic polymer coated ceramic fiber onto flat substrate. The process also includes the steps of using photo-light of the ultraviolet, visible, or infrared light spectrum to induce cross-linking (curing) of the photo-curable pre-ceramic polymer thereby rendering a thermoset polymer and either partially or completely pyrolyzing the now cured pre-ceramic polymer matrix coated ceramic fiber material. The pre-ceramic polymer poly(ethynyl)carbosilane yields silicon carbide upon pyrolysis. The pre-ceramic polymer may also yield oxide ceramic such as aluminum oxide upon pyrolysis. Other photo-curable pre-ceramic polymers may yield silicon nitride, aluminum nitride and titanium carbide, for example.

[0107] Various alternative methods for synthesizing a photo-curable pre-ceramic polymer are set forth below. A first process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of reacting sodium acetylide with organo-chlorosilanes and condensing (polymerizing) the resultant organo-(ethynyl)chlorosilane product of step a with an excess of an alkali metal. The organochlorosilane is may be selected from a group of one or more of the following: dichlorodimethylsilane, trichloro-phenylsilane (tri-functional), and methyltrichlorosilane.

[0108] A second process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of reacting sodium acetylide with organochloro-silanes and condensing (polymerizing) the resultant organo(ethynyl)-chlorosilane product of step a with an excess of an alkali metal sodium. The

organochlorosilane may be selected from a group of one or more of the following: dichlorodimethylsilane, trichloro-phenylsilane (tri-functional), and methyltrichlorosilane.

[0109] A third process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane, to silicon carbide ceramic includes the steps of reacting sodium acetylide with a mixture of organodichlorosilanes and organotrichlorosilanes and condensing (polymerizing) the resultant organo(ethynyl)-chlorosilane product of step a with an excess of an alkali metal.

[0110] A fourth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of reacting a sub-stoichiometric amount of an alkali metal with organochloro-silanes and reacting the partially polymerized polyorganochlorosilane with sodium acetylide. The organochlorosilane is may be selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0111] A fifth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of reacting a sub-stoichiometric amount of sodium metal with organochlorosilanes and reacting the partially polymerized polyorganochlorosilane with sodium acetylide. The organochlorosilane may be selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0112] A sixth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)carbosilane to silicon carbide ceramic includes the steps of reacting a sub-stoichiometric amount of an alkali metal with a mixture of organodichlorosilanes and organotrichlorosilanes and reacting the partially polymerized polyorganochlorosilane with sodium acetylide. The organochlorosilane may be selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0113] A seventh process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)silazane, to silicon nitride ceramic includes the steps of reacting sodium acetylide with organochlorosilanes and condensing (polymerizing) the resultant organo(ethynyl)chlorosilane product of step a with ammonia. The organochlorosilane may be selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0114] An eighth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of reacting sodium acetylide with organochlorosilanes and condensing (polymerizing) the resultant organo(ethynyl)-chlorosilane product of step a with ammonia.

[0115] A ninth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)silazane, to silicon nitride ceramic includes the steps of reacting sodium acetylide with a mixture of organodichlorosilanes and organotrichloro-silanes and condensing (polymerizing) the resultant organo(ethynyl)chlorosilane product of step a with ammonia. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0116] A tenth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of reacting a sub-stoichiometric amount of ammonia with organochlorosilanes and reacting the partially polymerized polyorganochlorosilazane with sodium acetylide. The organochlorosilane may be selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0117] An eleventh process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of reacting a sub-stoichiometric amount of ammonia with organochlorosilanes and reacting the partially polymerized polyorganochlorosilazane with sodium acetylide. The organochlorosilane

may be selected from a group consisting of one or more of the following:
dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0118] A twelth process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of reacting a sub-stoichiometric amount of ammonia with ~~with~~ a mixture of organodichlorosilanes and organotrichlorosilanes and reacting the partially polymerized polyorganochlorosilazane with sodium acetylide.

[0119] A thirteen process for fabricating a ceramic matrix composites includes the steps of preparing a solution of thermoplastic photo-curable pre-ceramic polymer, passing a pre-preg through the solution of thermoplastic photo-curable pre-ceramic polymer, applying the pre-preg to a shaped mandrel, using light energy to induce cross-linking of the photo-curable pre-ceramic polymer after application to the mandrel whereby the thermoplastic pre-ceramic polymer is curved and pyrolyzing the cured thermoplastic pre-ceramic polymer matrix composite material.

[0120] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of (a) reacting sodium acetylide with organo-chlorosilanes and (b) condensing (polymerizing) the resultant organo-(ethynyl)chlorosilane product of step a with an excess of an alkali metal. The organochlorosilane is selected from a group of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0121] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of (a) reacting sodium acetylide with organochlorosilanes and (b) condensing (polymerizing) the resultant organo(ethynyl)chlorosilane product of step a with an excess of an alkali metal sodium. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0122] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane, to silicon carbide ceramic includes the steps of (a) reacting sodium acetylide with a mixture of organodichlorosilanes and organotrichlorosilanes and (b) condensing (polymerizing) the resultant organo(ethynyl)-chlorosilane product of step a with an excess of an alkali metal.

[0123] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of (a) reacting a sub-stoichiometric amount of an alkali metal with organochlorosilanes and (b) reacting the partially polymerized polyorganochlorosilane with sodium acetylide. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0124] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-carbosilane to silicon carbide ceramic includes the steps of (a) reacting a sub-stoichiometric amount of sodium metal with organochlorosilanes and (b) reacting the partially polymerized polyorganochlorosilane with sodium acetylide. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane.

[0125] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)carbosilane to silicon carbide ceramic includes the steps of (a) reacting a sub-stoichiometric amount of an alkali metal with a mixture of organodichlorosilanes and organotrichlorosilanes and (b) reacting the partially polymerized polyorganochlorosilane with sodium acetylide.

[0126] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)silazane, to silicon nitride ceramic includes the steps of (a) reacting sodium acetylide with organochlorosilanes and (b) condensing (polymerizing) the resultant organo(ethynyl)chlorosilane product of step a with ammonia. The organochlorosilane is

selected from a group consisting of one or more of the following:

~~dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane~~

[0127] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of (a) reacting sodium acetylide with organochlorosilanes and (b) condensing (polymerizing) the resultant organo(ethynyl)chloro-silane product of step a with ammonia. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane

[0128] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)silazane, to silicon nitride ceramic includes the steps of (a) reacting sodium acetylide with a mixture of organodichlorosilanes and organotrichlorosilanes and (b) condensing (polymerizing) the resultant organo(ethynyl)chloro-silane product of step a with ammonia.

[0134] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)silazane to silicon nitride ceramic includes the steps of (a) reacting a sub-stoichiometric amount of ammonia with organochlorosilanes and (b) reacting the partially polymerized polyorganochlorosilazane with sodium acetylide. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane

[0129] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of (a) reacting a sub-stoichiometric amount of ammonia with organochlorosilanes and (b) reacting the partially polymerized polyorganochlorosilazane with sodium acetylide. The organochlorosilane is selected from a group consisting of one or more of the following: dichlorodimethylsilane, trichlorophenylsilane (tri-functional) and methyltrichlorosilane

[0130] A process of forming a photo-curable pre-ceramic polymer, poly(ethynyl)-silazane to silicon nitride ceramic includes the steps of (a) reacting a sub-stoichiometric amount of ammonia with a mixture of organodichlorosilanes and organotrichlorosilanes and (b) reacting the partially polymerized polyorganochlorosilazane with sodium acetylide.

[0131] A process for fabricating [a] ceramic matrix composites includes the steps of (a) preparing a solution of thermoplastic photo-curable pre-ceramic polymer, (b) passing a pre-preg through the solution of thermoplastic photo-curable pre-ceramic polymer, (c) applying the pre-preg to a shaped mandrel, (d) using light energy to induce cross-linking of the photo-curable pre-ceramic polymer after application to the mandrel whereby the thermoplastic pre-ceramic polymer is curved and (e) pyrolyzing the cured thermoplastic pre-ceramic polymer matrix composite material.

[0132] A first single step For further example, fabrication of continuous ceramic fiber ceramic matrix composites employing a thermoplastic photo-curable pre-ceramic polymer in which the component is shape by may include a variety of standard composite fabrication techniques, such as filament winding, tape winding, and woven cloth winding. An exemplary process includes steps of (a) passing ceramic fiber monofilament, tow, mat, or woven cloth through a solution of the thermoplastic photo-curable pre-ceramic polymer, (b) applying ceramic fiber monofilament, tow, mat, or woven cloth to a shaped mandrel, (c) using photo-energy of the ultraviolet, visible or infrared light spectrum to induce cross-linking (curing) of the photo-curable pre-ceramic polymer after application to the mandrel and (d) either partially or completely pyrolyzing the now cured pre-ceramic polymer matrix composite material. The pre-ceramic polymer is may comprise poly(ethynyl)carbosilane. The pre-ceramic polymer may yield silicon carbide upon pyrolysis. The pre-ceramic polymer may yield an oxide ceramic upon pyrolysis. The pre-ceramic polymer may yield titanium carbide upon pyrolysis. The pre-ceramic polymer may yield aluminum nitride upon pyrolysis. The pre-ceramic polymer may yield silicon nitride upon pyrolysis. The pre-ceramic polymer may yield aluminum oxide upon pyrolysis. In addition, or in the alternative, after pyrolysis the process may yield

titanium carbide, aluminum nitride, silicon nitride, or aluminum oxide, depending on the particular pre-ceramic polymer employed.

[0133] ~~A second single step fabrication of continuous ceramic fiber ceramic matrix composites employing a thermoplastic photo-curable pre-ceramic polymer in which the component is shape by a variety of standard composite fabrication techniques, such as filament winding, tape winding, and woven cloth winding under inert atmosphere includes~~ A second exemplary process includes steps of (a) passing ceramic fiber monofilament, tow, mat, or woven cloth through a solution of the thermoplastic photo-curable pre-ceramic polymer, (b) applying ceramic fiber monofilament, tow, mat, or woven cloth to a shaped rotating mandrel, (c) use of a heated or unheated compaction roller to press the thermoplastic pre-ceramic polymer onto the mandrel, (d) using ultraviolet, visible, or infrared light to induce cross-linking (curing) of the photo-curable pre-ceramic polymer thereby rendering a thermoset polymer, (e) either partially or completely pyrolyzing the now cured pre-ceramic polymer matrix material and (f) followed by the final heat treatment of the shaped ceramic matrix composite "brown body". The pre-ceramic polymer is may comprise poly(ethynyl)carbosilane. The pre-ceramic polymer may yield silicon carbide upon pyrolysis. The pre-ceramic polymer may yield an oxide ceramic upon pyrolysis. The pre-ceramic polymer may yield titanium carbide upon pyrolysis. The pre-ceramic polymer may yield aluminum nitride upon pyrolysis. The pre-ceramic polymer may yield silicon nitride upon pyrolysis. The pre-ceramic polymer may yield aluminum oxide upon pyrolysis. In addition, or in the alternative, after pyrolysis the process may yield titanium carbide, aluminum nitride, silicon nitride, or aluminum oxide, depending on the particular pre-ceramic polymer employed.

[0140] ~~A third single-step fabrication of continuous ceramic fiber ceramic matrix composites employing a thermoplastic photo-curable pre-ceramic polymer in which the component is shape by a variety of standard composite fabrication techniques, such as filament winding, tape winding, and woven cloth winding, includes steps of~~ (a) passing ceramic fiber monofilament, tow, mat, or woven cloth through a solution of the

thermoplastic photo-curable pre-ceramic polymer, (b) applying ceramic fiber monofilament, tow, mat, or woven cloth to a moving flat substrate, (c) using a compaction roller to press the thermoplastic pre-ceramic polymer coated ceramic fiber onto flat substrate, (d) using photo-light of the ultraviolet, visible, or infrared light spectrum to induce cross-linking (curing) of the photo-curable pre-ceramic polymer thereby rendering a thermoset polymer and (e) either partially or completely pyrolyzing the now cured pre-ceramic polymer matrix coated ceramic fiber material.[".] The pre-ceramic polymer is may comprise poly(ethynyl)carbosilane, yielding. ~~The pre-ceramic polymer may yield silicon carbide upon pyrolysis. The pre-ceramic polymer may yield an oxide ceramic upon pyrolysis. The pre-ceramic polymer may yield titanium carbide upon pyrolysis. The pre-ceramic polymer may yield aluminum nitride upon pyrolysis. The pre-ceramic polymer may yield silicon nitride upon pyrolysis. The pre-ceramic polymer may yield aluminum oxide upon pyrolysis. In addition, or in the alternative, after pyrolysis the process may yield titanium carbide, aluminum nitride, silicon nitride, aluminum oxide, or other oxide ceramic, depending on the particular pre-ceramic polymer employed~~

[0134] As illustrated by the following example, photocurable Photocurable poly(ethynyl)carbosilane can be synthesized directly from difunctional and trifunctional chlorosilane reagents with the addition of sub-stoichiometric amounts of sodium to form poly(chloro) silanes, followed by the addition of excess sodium acetylide to provide photocurable cross-linking sites.

[0135] Sodium metal suspension (40% by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. The mixture of methylene bromide, dichlorodimethylsilane, and trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The

mixture was stirred for an additional hour while cooling. The dark purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0136] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in tetrahydrofuran (THF). The appropriate amount of sodium acetylide powder was dissolved in dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution and an exothermic reaction occurs and the color of the polymer solution turned a deep orange. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer was obtained.

[0137] Six different examples of PECS (poly(ethynyl)carbosilane), with varying ethynyl groups concentrations have been prepared as shown in Table 4. Ethynyl concentration was varied from 0 to 25 percent (by mole).

[0138] In order to characterize the molecular weight and molecular weight distributions of polymers synthesized and utilized in this study, HPLC was utilized. A carefully prepared calibration curve was measured using NIST traceable molecular weight standards and measuring elution time. From this calibration curve, we were able to estimate the peak molecular weight of the PECS synthesized based upon the chromatograms.

[0139] TABLE 4: Poly(ethynyl)carbosilane polymers synthesized in this study.

Poly(ethynyl)carbosilane Synthesis Matrix												
	Reactants		TRIAL 1			TRIAL 2			TRIAL 3			
			0%A/100%B			5%A/95%B			10%A/90%B			
	NAME		MW	MOL	ML	GM	MOL	ML	GM	MOL	ML	GM
A	TCMS		149.48	0.0000	0.0000	0.000	0.0025	0.2936	0.374	0.0050	0.5871	0.747
B	DCMS		129.07	0.0500	6.0653	6.454	0.0475	5.7621	6.131	0.0450	5.4588	5.808
C	Methylene		173.85	0.0500	3.5093	8.693	0.0500	3.5093	8.693	0.0500	3.5093	8.693
D	NA		23	0.2000		4.600	0.2000		4.600	0.2000		4.600
E	NA		48	0.0000		0.000	0.0025		0.120	0.0050		0.240
D	40% Na-Oil					11.500			11.500			11.500

E	17.3% Na				0.000			0.694			1.387
			TRIAL 3			TRIAL 4			TRIAL 5		
			15%A/85%B				20%A/80%B			25%A/75%B	
	NAME	MW	MOL	ML	GM	MOL	ML	GM	MOL	ML	GM
A	TCMS	149.48	0.0075	0.8807	1.121	0.0100	1.1742	1.495	0.0125	1.4678	1.869
B	DCMS	129.07	0.0425	5.1555	5.485	0.0400	4.8523	5.163	0.0375	4.5490	4.840
C	Methylene	173.85	0.0500	3.5093	8.693	0.0500	3.5093	8.693	0.0500	3.5093	8.693
D	NA	23	0.2000		4.600	0.2000		4.600	0.2000		4.600
E	NA	48	0.0075		0.360	0.0100		0.480	0.0125		0.600
D	40% Na-Oil				11.500			11.500			11.500
E	17.3% Na				2.081			2.775			3.468

[0140] In Table 5 below, several of our polymers are compared with Dow Corning PCS. Our materials were purposely prepared as viscous fluids for greater ease in fabrication.

[0141] TABLE 5: Molecular Weights and HPLC Elution Times (peak) for PECS Synthesized by MATECH and Compared with Dow Corning PCS

POLYMER	ELUTION TIME	MORPHOLOGY	MOLECULAR WEIGHT
Dow Corning PCS	14.468	Solid Flake	4400
PECS (0% ethynyl) A	16.598	Viscous Fluid	750
PECS (0% ethynyl) B	16.449	Viscous Fluid	700
PECS (5% ethynyl)	16.050	Viscous Fluid	1300
PECS (15% ethynyl)	16.862	Viscous Fluid	600
PECS (20% ethynyl) A	16.504	Viscous Fluid	700
PECS (20% ethynyl) B	16.973	Viscous Fluid	1400
PECS (25% ethynyl)	16.732	Viscous Fluid	580

[0142] One of the polymers synthesized as described above was used to fabricate a ceramic matrix composite using woven ceramic fabric. 7.0 grams of Poly(ethynyl)carbosilane with 15% ethynyl side-groups for cross-linking was impregnated into 4 layers of woven ALTEX fabric. The resulting pre-preg was

photocured over night to produce cross-linked matrix and then fired in Argon gas to 1200 °C for one hour. The resulting product was a ceramic coupon suitable for testing and evaluation.

[0143] The polymer synthesized above, 7.0 grams of Poly (ethynyl)carbosilane with 15% ethynyl side-groups for cross-linking was impregnated into 4 layers of woven ALTEX fabric. The resulting pre-preg was photocured overnight to produce cross-linked matrix and then fired in Argon gas to 1200 °C for one hour. The resulting product was a ceramic coupon suitable for testing and evaluation.

[0144] The resulting SiC ceramic matrix composite (CMC) has been characterized. After only two processing cycles, the resulting CMC has an apparent density of 2.134 grams/cc only two processing cycles, the resulting CMC has an apparent density of 2.134 grams/cc and a porosity of 38.24 percent (%). In addition, it exhibits good strength and sounds very much like a ceramic when tapped. Scanning Electron Microscopy (SEM) photomicrographs reveal that the woven fiber tows (of approximately 500 monofilaments each) are well bonded with minimal porosity, even at high magnification. Large pores are still present between tows, however, which can permit further densification through repeated polymer-impregnation-pyrolysis (PIP) cycles.

EXAMPLE 10

[0145] For 25% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 4.840 grams dichlorodimethylsilane, and 1.869 grams trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark

purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0146] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in 50 ml tetrahydrofuran (THF). 0.600 grams of sodium acetylide powder was dissolved in 5.0 ml dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution. An and an exothermic reaction occurred and the color of the polymer solution turned a deep purple-red. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer dissolved in THF was obtained. The polymer was then extracted from the filtrate by evaporation in a Rotovapor apparatus, yielding approximately 8.0 grams of poly(ethynyl)carbosilane.

EXAMPLE 11

[0147] For 20% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 5.163 grams dichlorodimethylsilane, and 1.495 grams trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0148] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in 50 ml tetrahydrofuran (THF). 0.480 grams of sodium acetylide powder was dissolved in 5.0 ml dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution and an exothermic reaction occurred and the

color of the polymer solution turned a deep purple-red. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer dissolved in THF was obtained. The polymer was then extracted from the filtrate by evaporation in a Rotovapor apparatus, yielding approximately 8.0 grams of poly(ethynyl)carbosilane.

EXAMPLE 12

[0149] For 15% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 5.485 grams dichlorodimethylsilane, and 1.121 grams trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0150] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in 50 ml tetrahydrofuran (THF). 0.360 grams of sodium acetylide powder was dissolved in 5.0 ml dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution and an exothermic reaction occurred and the color of the polymer solution turned a deep purple-red. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer dissolved in THF was obtained. The polymer was then extracted from the filtrate by evaporation in a Rotovapor apparatus, yielding approximately 8.0 grams of poly(ethynyl)carbosilane.

EXAMPLE 13

[0151] For 10% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in

xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 5.808 grams dichlorodimethylsilane, and 0.747 grams trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0152] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in 50 ml tetrahydrofuran (THF). 0.240 grams of sodium acetylide powder was dissolved in 5.0 ml dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution and an exothermic reaction occurred and the color of the polymer solution turned a deep purple-red. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer dissolved in THF was obtained. The polymer was then extracted from the filtrate by evaporation in a Rotovapor apparatus, yielding approximately 8.0 grams of poly(ethynyl)carbosilane.

EXAMPLE 14

[0153] For 5% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 6.131 grams dichlorodimethylsilane, and 0.374 grams trichlorophenylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark

purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0154] The resulting poly(chloro)carbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus. The resulting dark yellow viscous polymer was dissolved in 50 ml tetrahydrofuran (THF). 0.120 grams of sodium acetylide powder was dissolved in 5.0 ml dimethyl formamide (DMF) and added slowly to the poly(chloro)carbosilane polymer solution and an exothermic reaction occurred and the color of the polymer solution turned a deep purple-red. Reaction byproducts were removed by filtration and the final poly(ethynyl)carbosilane polymer dissolved in THF was obtained. The polymer was then extracted from the filtrate by evaporation in a Rotovapor apparatus, yielding approximately 8.0 grams of poly(ethynyl)carbosilane.

EXAMPLE 15

[0160] For 0% ethynyl side-group substitution, 11.50 grams of sodium metal suspension (40% sodium by weight) in oil was weighed. The suspension was washed three times in xylene and separated by centrifugation. The washed sodium was added to 200 ml of xylene in the triple-neck reaction vessel. The refluxed reaction vessel was heated under flowing argon to 100 °C. A mixture of 8.693 grams methylene bromide, 6.454 grams dichlorodimethylsilane was slowly added using a burette. An exothermic reaction ensued and the temperature of reaction vessel contents reached 133 °C and the mixture boiled vigorously under reflux for approximately 30 minutes. The mixture was stirred for an additional hour while cooling. The dark purple/brown mixture, containing precipitates, was filtered and a clear yellow filtrate was obtained.

[0155] The resulting polycarbosilane polymer was extracted from the filtrate by evaporation in a Rotovapor apparatus yielding approximately 8.0 grams of polycarbosilane with no ethynyl side-groups.

[0156] As demonstrated by the following examples, It has been demonstrated that several commercially available preceramic polymers can be made photocurable. The preceramic

polymer CERASETTM SZ inorganic polymer sold by Honeywell Advanced Composites, Inc., which is a silazane-based polymer, can be made photocurable to both UV and blue light through the addition of photoinitiators. Also, the preceramic polymer allylhydridopolycarbosilane (AHPCS) polymer manufactured by Starfire Systems, Inc. can be made photocurable to both UV and blue light through the addition of photoinitiators.

EXAMPLE 16

[0157] A UV light photocurable polysilazane was produced by mixing 2.00 grams of CERASETTM SZ inorganic polymer with 0.50 grams of IRGACURE® 1800, manufactured by Ciba Specialty Chemicals, dissolved in 0.50 ml tetrahydrofuran. The resulting yellow fluid, upon exposure to a high intensity UV lamp, became a stiff, rigid polymer within an hour. The resulting cross-linked polymer maintained its shape upon heating and pyrolysis to 1200 °C in flowing argon gas. The ceramic yield of the pyrolyzed polymer was in excess of 80 percent. A control sample, without the photoinitiator, remained fluid after in excess of 24 hours of continuous UV irradiation.

EXAMPLE 17

[0158] A blue light photocurable polysilazane was produced by mixing 2.00 grams of CERASETTM SZ inorganic polymer with 0.50 grams of Camphorquinone, obtained from Aldrich Chemical Company, dissolved in 0.50 ml tetrahydrofuran. The resulting yellow fluid, upon exposure to a high intensity blue lamp, became a stiff, rigid polymer within an hour. The resulting cross-linked polymer maintained its shape upon heating and pyrolysis to 1200 °C in flowing argon gas. The ceramic yield of the pyrolyzed polymer was in excess of 80 percent. A control sample, without the photoinitiator, remained fluid after in excess of 24 hours of continuous blue light irradiation.

EXAMPLE 18

[0159] A UV light photocurable allylhydridocarbosilane was produced by mixing 2.00 grams of allylhydridocarbosilane (15% allylchloride) polymer with 0.50 grams of IRGACURE® 1800, manufactured by Ciba Specialty Chemicals, dissolved in 0.50 ml

tetrahydrofuran. The resulting yellow fluid, upon exposure to a high intensity UV lamp, became a stiff, rigid polymer within an hour. The resulting cross-linked polymer maintained its shape upon heating and pyrolysis to 1200 °C in flowing argon gas. The ceramic yield of the pyrolyzed polymer was in excess of 80 percent. A control sample, without the photoinitiator, remained fluid after in excess of 24 hours of continuous UV irradiation.

EXAMPLE 19

[0160] A blue light photocurable allylhydridocarbosilane was produced by mixing 2.00 grams of allylhydridocarbosilane (15% allylchloride) polymer with 0.50 grams of Camphorquinone, obtained from Aldrich Chemical Company, dissolved in 0.50 ml tetrahydrofuran. The resulting yellow fluid, upon exposure to a high intensity blue lamp, became a stiff, rigid polymer within an hour. The resulting cross-linked polymer maintained its shape upon heating and pyrolysis to 1200 °C in flowing argon gas. The ceramic yield of the pyrolyzed polymer was in excess of 80 percent. A control sample, without the photoinitiator, remained fluid after in excess of 24 hours of continuous blue light irradiation.

[0161] From the foregoing it can be seen that processes of forming a photo-curable pre-ceramic polymer and their applications have been described.

[0162] Accordingly it It is intended that the foregoing disclosure shall be considered only as an illustration of the principle of the present process. The invention is defined by the appended claims.